

Switchable Plasmonic-Dielectric Resonators with Metal-Insulator Transitions

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Abstract

Nanophotonic resonators offer the ability to design nano-scale optical elements and engineered materials with unconventional properties. Dielectric-based resonators intrinsically support a complete multipolar resonant response with low absorption, while metallic resonators provide extreme light-confinement and enhanced photon-electron interactions. Here, we construct resonators out of a prototypical metal-insulator transition material, Vanadium Dioxide (VO₂), and demonstrate switching between dielectric and plasmonic resonances. We first characterize the temperature-dependent infrared optical constants of VO₂ single-crystals and thin-films. We then fabricate VO₂ wire arrays and disk arrays. We show that wire resonators support dielectric resonances at low temperatures, a damped scattering response at intermediate temperatures, and plasmonic resonances at high temperatures. In disk resonators, however, upon heating there is a pronounced enhancement of scattering at intermediate temperatures, and a substantial narrowing of the phase transition. These findings may lead to the design of novel nanophotonic devices that incorporate thermally switchable plasmonic-dielectric behavior.

Keywords: tunable metasurfaces, metal-insulator transition, phase change materials, vanadium dioxide

Nanophotonic resonators have made an enormous impact on nanoscale science and engineering. Metallic resonators that support plasmonic excitations have enabled extreme concentration of light¹, near-field imaging microscopes that surpass the diffraction limit², plasmon-driven catalytic processes^{3,4}, antennas with localized heat generation⁵, and metamaterials that support hyperbolic optical response⁶. The more recent development of dielectric Mie-type resonators, which intrinsically support a multipolar response with both electric and magnetic modes, have further expanded the available parameter space for nanophotonic design⁷⁻⁹. Dielectric resonators are used for a wide range of functionalities including evanescent wave skin-depth engineering¹⁰, amplification of non-linear processes¹¹, and local enhancement of magnetic fields^{12,13}. Furthermore, dielectric metasurfaces offer the potential ability to replace large traditional optics with high efficiency, compact, nano-scale optical elements. By spatially varying the geometry across a surface, flat optical components were recently demonstrated, including lenses^{14,15,16}, blazed gratings^{17,18,19}, vortex plates^{20,21}, and beam-shaping metasurfaces^{22,23}.

There is substantial interest in developing nanophotonic devices with dynamically reconfigurable properties²⁴. To satisfy this need, various physical mechanisms have been pursued, including electrical carrier injection/depletion of semiconductors^{25,26}, amorphous-crystalline phase changes in dielectrics^{27,28,29}, nano-mechanical tuning of resonators³⁰, ultrafast pumping^{31,32}, thermo-optic tuning³³, voltage gating of graphene³⁴, and metal-insulator transitions (MIT) in strongly correlated materials^{35,36,37,38,39,40}. In this work, we demonstrate active index-tunable resonators comprising the prototypical strongly correlated material, vanadium dioxide (VO₂). Exploiting a temperature-driven MIT, we demonstrate large shifts in the multipolar resonant behavior as resonators switch between plasmonic and dielectric regimes. In contrast to previous work on VO₂ optical elements^{41,42,43,44}, these metamaterial resonators support switching between dielectric Mie resonances and plasmonic resonances. We grow single crystals and thin films of VO₂ and characterize the infrared optical permittivity across the MIT. We then fabricate and experimentally characterize VO₂ wire arrays and disk arrays. We find that VO₂ wire arrays exhibit a polarization-dependent dielectric resonance at low temperatures, a suppressed scattering response at intermediate temperatures, and a plasmonic resonance at high temperatures. The VO₂ disk arrays on the other hand, exhibit enhanced scattering at intermediate temperatures and a narrowed metal-insulator transition. This proof-of-concept

paves the way towards constructing thermally tunable metasurfaces that integrate switchable plasmonic-dielectric behavior.

Tunable Optical Constants

Strongly correlated materials are substances in which strong electron-electron interactions lead to emergent properties such as superconductivity, colossal magnetoresistance, and Mott-Hubbard metal-insulator phase transitions⁴⁵. When heat, voltage, or magnetic fields are applied to such materials the optical constants may also undergo dramatic shifts. In Fig. 1a, we plot the reflectivity of a VO₂ single crystal in its insulating and metallic phases, and, in Fig. 1b, we plot the fitted optical permittivity. The insulating phase is characterized by a flat reflection spectrum with $R \approx 0.25$, corresponding to a permittivity $\epsilon \approx 9$ (refractive index $n \approx 3$), slightly lower than that of Silicon ($n \approx 3.4$). The metallic phase is characterized by a dispersive reflection spectrum typical of low conductivity metals, which corresponds to a Drude permittivity model with plasma frequency $\omega_p = 413$ THz and scattering rate $\gamma = 74$ THz.

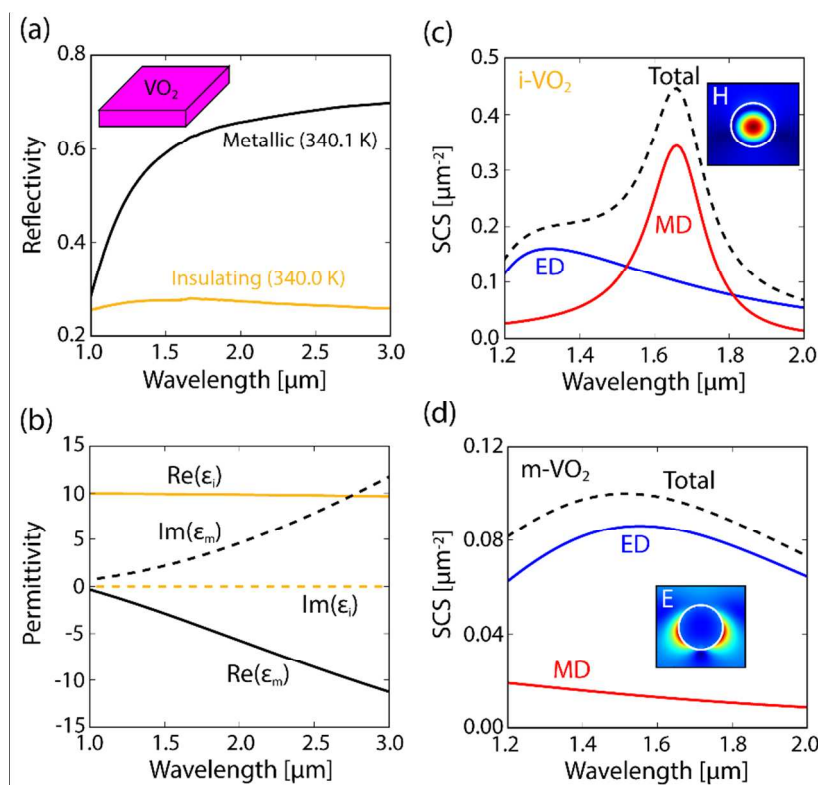


Figure 1 – Reflectivity of VO₂ Single Crystals. (a) Near-infrared reflectivity of a VO₂ single crystal in its low-temperature insulating phase (orange curve) and high-temperature metallic phase (black curve). (b) Real (solid curve) and imaginary (dashed curve) components of the fitted VO₂ optical permittivity in the insulating phase (orange curve) and metallic phase (black curve). Scattering-cross-section of a (c) insulating, and (d) metallic VO₂ sphere in air, with diameter $d=500$ nm, calculated with Mie theory. Insets corresponds to simulated local (c) magnetic, and (d) electric, field profile at the fundamental resonance. Crystal growth, optical characterization, Mie Theory calculation, and simulation details are included in the Supporting Information.

The contrast in reflectivity and permittivity is most pronounced at wavelengths beyond $1.5 \mu\text{m}$. We plot the scattering-cross-section of a 500 nm VO₂ sphere, calculated with Mie theory, in the insulating (Fig. 1c) and the metallic (Fig. 1d) phases. In both phases, a fundamental scattering resonance occurs near $1.65 \mu\text{m}$. However, this resonance is primarily a magnetic-dipole mode in the insulating phase and an electric-dipole mode in the metallic phase. The electric and magnetic profiles are shown in insets of Fig. 1c and 1d. Such resonators can therefore be used as the basis for optical antennas that switch from magnetic to electric resonant behavior. In the Supporting

Information we experimentally investigate the behavior of individual Vanadium Oxide spherical resonators, fabricated with femtosecond laser ablation.

In free-standing single crystals the metal-insulator transition is sharp and characterized by phase boundary propagation from few nucleation sites⁴⁶. In thin-films, a more common medium for nanophotonic resonators, defects generate many nucleation sites, leading to a broadened percolative transition⁴⁷. As the insulating film is heated, nanoscale metallic inclusions nucleate and percolate into a fully metallic film⁴⁸. Optically, the film acts as an effective medium, providing continuous access to intermediate states with optical constants averaged between the insulating and the metallic clusters. In Fig. 2a, we plot the infrared reflectivity of a 100 nm thick VO₂ film on a sapphire substrate at various temperatures. In the insulating phase, the film exhibits a Fabry-Perot resonance near $\lambda=4nt=1.2\ \mu\text{m}$, where n is the refractive index and t is the film thickness. Note that this is a reflection *maximum* due to the larger index of refraction in VO₂ in comparison to the sapphire substrate. As the temperature increases the feature smears out due to increased losses, culminating in a near flat response at an intermediate temperature of 341K. Although intermediate values of $\text{Re}[\epsilon]$ are accessed at intermediate temperatures (Fig. 2b), the sharp rise in losses (Fig. 2c) precludes any clear shifting of the Fabry-Perot resonance wavelength. There is also a local minimum in $\text{Re}[\epsilon]$ due to the presence of an absorption resonance. This local minimum is not present in single-crystals and might be due to defects in the thin film. At higher temperatures, metallic behavior ultimately sets in.

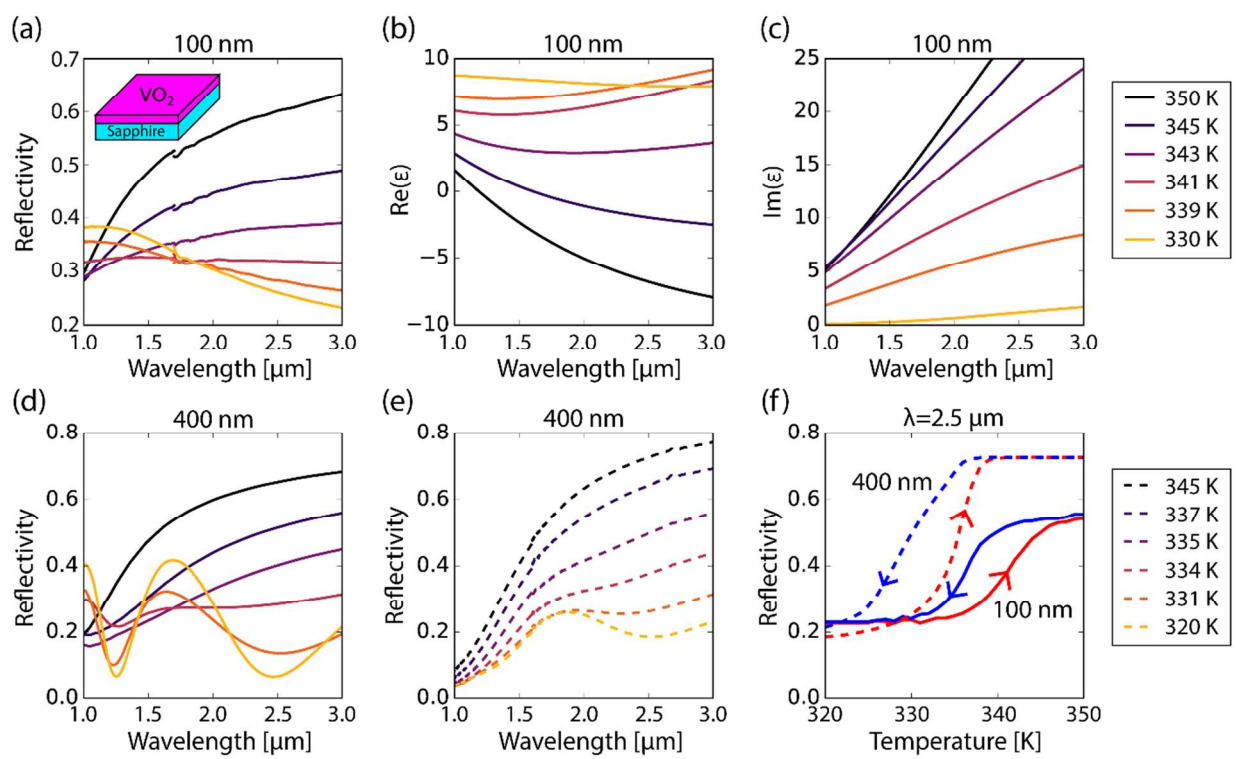


Figure 2 – VO₂ Thin-Film Reflectivity and Optical Constants (a) The measured FTIR reflectivity of a 100 nm VO₂ film on a sapphire substrate. The (b) real, and (c) imaginary components of the optical permittivity, as determined by a spectral fit to the measured reflectivity. Experimental and fitting details are provided in the Supporting Information. (d) The simulated reflectivity of a 400 nm thick VO₂ film, based on the 100 nm optical constants. (e) The measured reflectivity of a 400 nm thick VO₂ film. (f) The optical hysteresis of the 100 nm film (solid line) and 400 nm film (dashed line) at a wavelength $\lambda=2.5\ \mu\text{m}$. Sample fabrication, measurement details, and mid-infrared spectra are included in the Supporting Information.

As the film is heated through the MIT, the initially positive permittivity, $\epsilon \approx 9$, continuously decreases, passes through zero, and becomes negative, corresponding to the metallic phase. When fully metallic, the Drude fit has a plasma frequency $\omega_p = 981\ \text{THz}$ and scattering rate $\gamma = 205\ \text{THz}$. The larger scattering rate, as compared to single

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3 crystals, may be attributed the larger number of defects in the films in the form of grain boundaries, crystallographic
4 defects, and surface roughness. This full span of permittivity, corresponding to capacitive (dielectric), inductive
5 (metallic), and resistive (intermediate) optical behavior, signifies that VO₂ may be well-suited for the design of
6 reconfigurable devices based on optical circuit concepts⁴⁹.
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8 In order to support high-quality dielectric-type resonances in the infrared spectral range, where permittivity
9 contrast is large, the VO₂ film must be relatively thick. In Fig. 2d, we plot the simulated reflectivity of a 400 nm
10 thick VO₂ film deposited on a sapphire substrate based on the extracted optical constants of the 100 nm film. In the
11 fully insulating low-temperature phase (yellow), we observe a series of higher-order Fabry-Perot peaks
12 corresponding to $\lambda = 4nt/3 \approx 1.6\mu\text{m}$ and $\lambda = 4nt/5 \approx 1.0\mu\text{m}$. In the high-temperature metallic phase (black), the
13 simulated reflection again exhibits a typical Drude reflectivity profile. In Fig. 2e, we plot the experimentally
14 measured reflectivity of a 400 nm thick VO₂ film on a sapphire substrate. The observed peak near 1.6 μm in the
15 insulating phase, similar to the one seen in simulation, reveals a Fabry-Perot mode corresponding to a film refractive
16 index of 3. A subsequent Fabry-Perot reflection minimum is observed near 2.5 μm in both simulation and
17 experiment. The lack of an experimentally observed higher-order mode at 1 μm , however, suggests that there are
18 significant differences between the optical constants of thin and thick films. A good fitting of the thick film optical
19 constants could not be obtained due to increased surface roughness which suppresses specular reflection at shorter
20 wavelengths. Islands of non-stoichiometric VO_x, and strain-relaxation effects⁵⁰, may also contribute to differences
21 between simulated (based on thin-film constants) and measured results at short wavelengths. Compared to the 100
22 nm film, the thick-film MIT temperature is reduced by 5-7 K (Fig 2f) which may be attributed to oxygen
23 deficiency⁵¹. Additionally, the change in sheet resistance across the MIT is three orders of magnitude lower than the
24 100nm films (Fig. S4), indicating a clear difference in film quality.
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27 In subsequent simulations, we use the fitted optical constants of the 100 nm films to understand the
28 behavior of resonators that undergo metal-insulator phase transitions. This data is publicly available in our
29 supporting information. It is important to note that between films there may be variations in the reflectivity and,
30 hence, optical constants due to differences in stoichiometry, substrate-induced strain, film thickness, grain size, and
31 surface roughness.
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33 Wire Arrays

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35 Semiconductor wires, when confined to sub-wavelength scales, support a series of multipolar Mie
36 resonances^{52,53,54}. Such structures are of interest for improved solar absorbers^{55,56}, photodetectors⁵⁷, and color
37 filters⁵⁸. Here we examine the behavior of a VO₂ wire array as it undergoes a metal-insulator transition. In Fig. 3a,
38 we plot the simulated reflectivity in the insulating phase (orange curve) and metallic phase (black curve). The
39 incident light is transverse-electric (TE) polarized with electric-field perpendicular to the wire orientation. The
40 fundamental dielectric resonance at $\lambda \approx 1.25\mu\text{m}$ has a strong magnetic-field inside the wire (inset of Fig. 3a) and is a
41 magnetic-dipole-like mode. The metallic resonance at $\lambda \approx 1.6\mu\text{m}$ on the other hand, is an electric-dipole-like mode.
42 Similar to spheres (Fig. 1c and d), wire resonators support both dielectric- and metallic-type resonances in the same
43 wavelength regime, but with different field symmetries. In the experimentally measured reflection spectrum (Fig.
44 3b) we observe similar reflection profiles and switching between magnetic and electric dipole resonances. The
45 experimentally measured reflection is reduced however, due to surface roughness. In Fig. 3c we plot the simulated
46 reflection spectra, including intermediate temperatures. At intermediate temperatures (dashed line near 340K), there
47 is a broadband suppression in scattering. This effect, similar to that observed in a previous work⁵⁹, may be attributed
48 to the relatively large loss tangent ($\text{Im}[\epsilon]/\text{Re}[\epsilon]$) when the VO₂ passes through its epsilon-near-zero condition, and
49 may be used to construct efficient amplitude modulators. Experimentally, we observe similar behavior in Fig. 3d
50 although decreased in amplitude, shifted to lower wavelength, and increased in the transition temperature. Similar
51 behavior is observed under transverse-magnetic polarization (supporting information).
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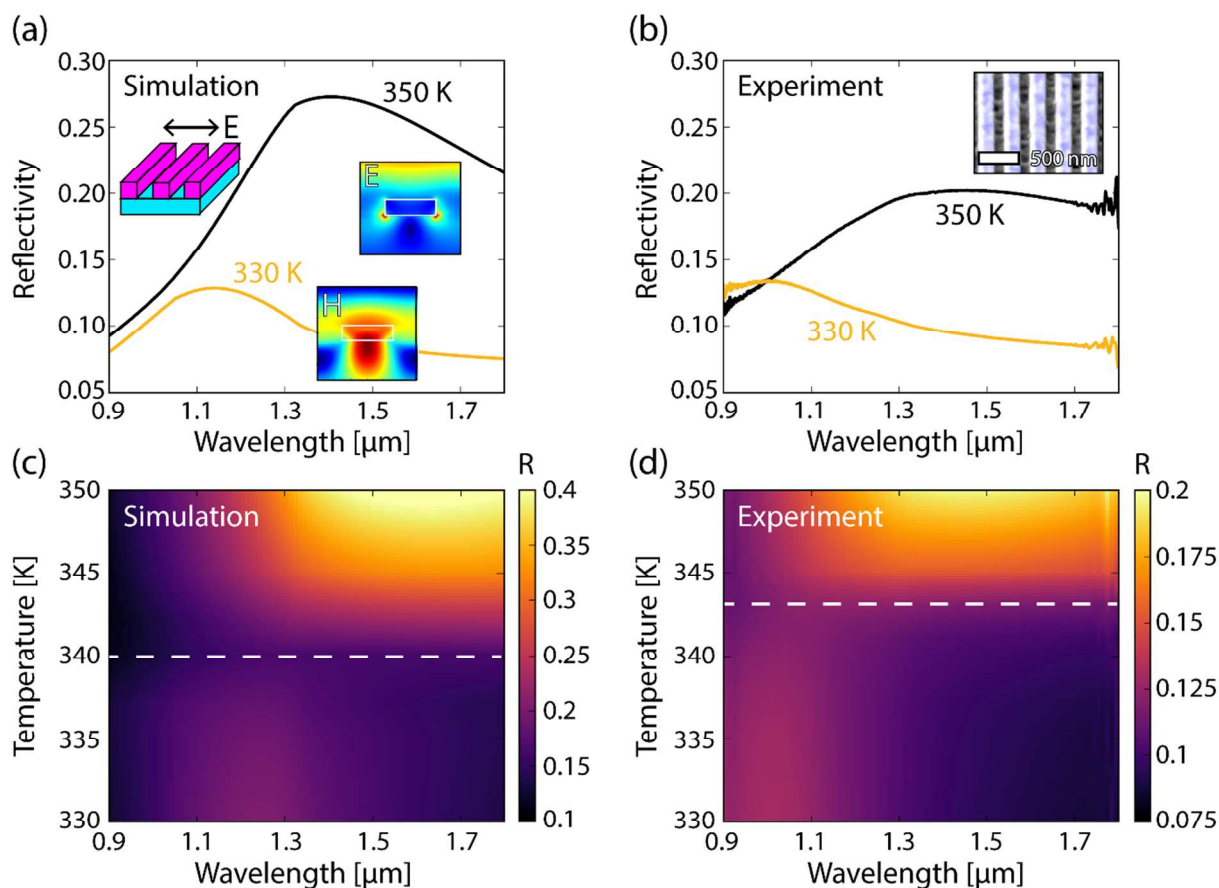


Figure 3 – **TE-Polarized VO₂ Wire Array Reflectivity.** (a) Simulated reflectivity of a VO₂ wire array, on sapphire, in the insulating and metallic phases, under TE-polarized illumination based on optical constants found for a 100 nm VO₂ film. Insets corresponding to the magnetic-field profiles at the insulating resonance, and electric-field profiles at the metallic resonance. (b) Experimentally measured reflectivity of the corresponding VO₂ wire array, in the insulating and metallic phases. Inset is an SEM of the fabricated structure. (c) Simulated, and (d) experimentally measured, reflectivity of the array, including intermediate temperatures. Note that the color scale in (d) is different relative to (c) to increase clarity. Wires are rectangular with width $w=280$ nm, height $h=100$ nm, and pitch $a=500$ nm. Simulation and experimental details are included in the Supporting Information.

Disk Arrays

VO₂ disk resonators are expected to exhibit more pronounced resonant features due to increased electromagnetic confinement. In Fig. 4a, we plot the simulated reflection spectra of a VO₂ disk array above and below the transition temperature. In the fully insulating phase (orange curve), the reflectivity peaks at $\lambda \approx 1.5$ μm, corresponding to the magnetic dipole resonance. As the array is heated across the transition, the resonance red-shifts and increases in amplitude, becoming a plasmonic electric dipole mode near $\lambda \approx 1.7$ μm (black curve). In Fig. 4b, we plot the complete simulated temperature-wavelength reflectivity dependence. Similar to the wire array, there is suppressed scattering at intermediate temperatures. In Fig. 4c, we plot the experimentally-measured reflection of a VO₂ disk array below and above the MIT. In the low-temperature insulating phase (orange curve) the measured response has a reflection peak at $\lambda \approx 1.5$ μm, in agreement with the simulated response. Upon heating, the array transitions from fully insulating behavior at 334 K to fully metallic behavior at 340 K, a substantially narrower transition than one found in the wire arrays or thin films. In the metallic phase (black curve), the measured response has a reflection peak at $\lambda \approx 1.3$ μm, slightly blue-shifted and with overall lower reflectivity value relative to the

simulation, which may be attributed to surface roughness and potentially direct contact between resonators. However, examining the experimentally-measured reflection, upon heating, (Fig. 4d) at intermediate temperatures there is a pronounced *enhancement* of reflection. Such behavior could be expected of a dielectric resonator with an increased refractive index, or a plasmonic resonator with an increased plasma frequency or reduced Drude scattering rate. It may also perhaps be attributable to strong-correlation effects in the vicinity of phase transition, which have been observed by others. Another potential explanation is the “critical opalescence” effect observed in inhomogeneous mixtures. Similar behavior was recently observed for VO₂ particle arrays in the visible spectrum⁶⁰. The effect was attributed to the random nature of the phase transition, in which individual particles randomly become metallic at different temperatures. This behavior is only present upon the heating portion of the hysteresis loop. Upon cooling (Fig. 4e) there is a suppression of reflection at intermediate temperatures, similar to the wire arrays. Such asymmetries may be attributable to the differing nucleation mechanisms upon heating and cooling⁶¹. Further studies are necessary to conclusively understand the origin of this enhanced scattering.

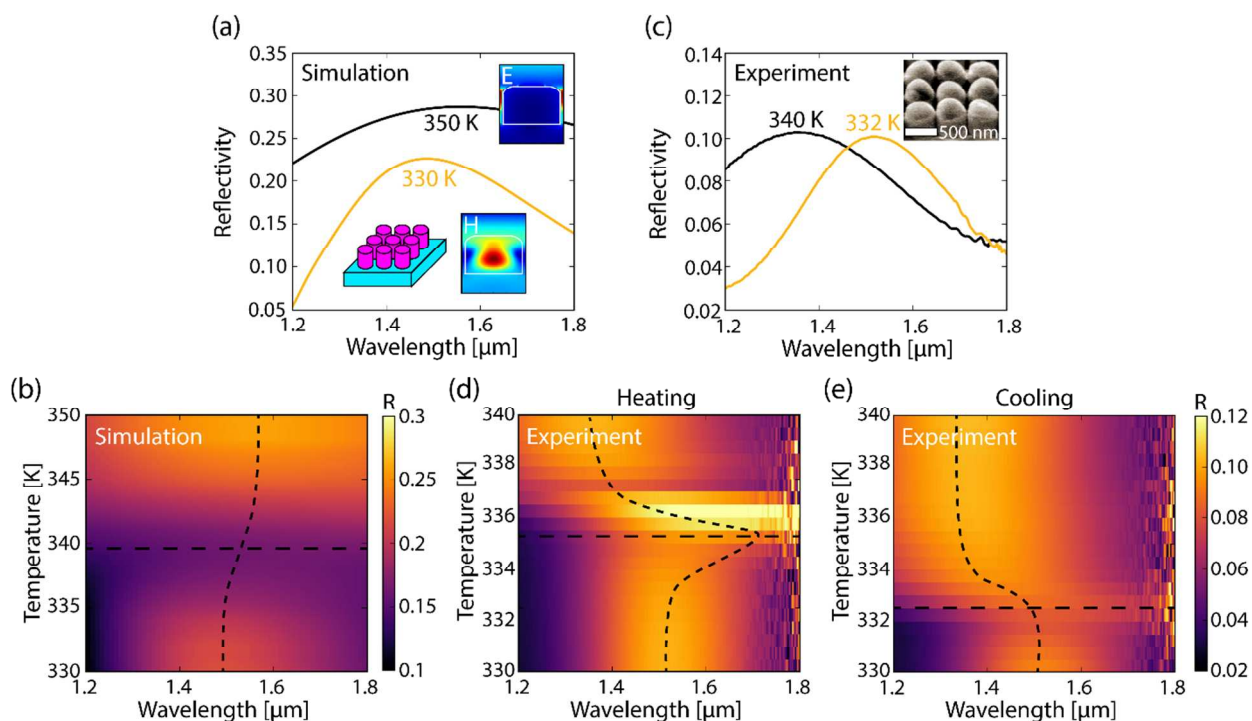


Figure 4 –VO₂ Disk Array Reflectivity. (a) Simulated reflectivity of a VO₂ disk array. Disk resonators have a height $h=500$ nm, diameter $d=480$ nm, and pitch $a=500$ nm based on optical constants from a 100 nm film. Insets correspond to local field distributions at the resonant peaks. (b) Simulated reflection as a function of temperature and wavelength. Vertical dashed curve tracks the reflectivity peak, horizontal dashed line represents the transition temperature (c) Experimentally measured reflectivity of VO₂ disk array (with approximate height $h=400$ nm, diameter $d=480$ nm, and pitch $a=500$) above and below transition. Inset is an SEM of the fabricated structure. (d) Experimental reflection data, upon heating, including intermediate temperatures. (e) Experimental reflection data, upon cooling, including intermediate temperatures. Vertical dashed curve tracks the reflectivity peak, horizontal dashed line represents the transition temperature. Note that the temperature span in (d,e) is different than (c).

Simulation and experimental details are included in the Supporting Information.

Future work on controlling the MIT of VO₂ nanophotonic devices, such as strain engineering^{62,63}, electrically controllable resonators^{64,65} and spatially-modulated optical pumping⁶⁶, may enable the development of high-performance reconfigurable metasurfaces. In previous works, tuning of statically-doped and photo-excited semiconductor resonators was demonstrated, in which dense electron-hole charge densities led to plasmonic optical behavior^{67,68,69,70}. VO₂ resonators, in contrast, display dynamic switching between dielectric and plasmonic behavior

over extremely narrow temperature spans, making them particularly intriguing for energy efficient applications. Finally, although here we demonstrated the basic behavior with VO₂, similar phenomena hold for other correlated materials such as V₂O₃^{71,72}, SmNiO₃⁷³, and NdNiO₃⁷⁴.

Conclusion

We demonstrate that the metal-insulator transition of a prototypical strongly correlated material, VO₂, may be used to construct thermally tunable structures that exhibit a dual plasmonic-dielectric resonant response. Through FTIR reflectivity measurements we extract the optical permittivity of VO₂ single crystals and thin-films above and below the MIT, allowing access to the full span of positive, zero, and negative optical permittivity. In fabricated arrays of VO₂ wires, we experimentally demonstrate dielectric resonances at low temperatures, a damped scattering response at intermediate temperatures, and a plasmonic resonance at high temperatures. Similar behavior occurs in disk resonators; however, increased confinement leads to several unusual effects. There is a substantial narrowing of the transition temperature width, such that switching occurs over less than 4 K. And there is a pronounced enhancement, rather than suppression, of scattering at intermediate temperatures. These novel, dynamically tunable optical behaviors of VO₂ resonators demonstrate that materials with metal-insulator phase transitions hold great promise for future meta-devices.

Supporting Information

Included in the Supporting Information are the experimental and simulation details, experimental data of wire arrays in TM-polarization, measurements of Vanadium oxide spheres, measurements of randomly arranged disk resonators, a Mathematica notebook containing the code used for Mie Theory calculations, and the extracted optical constants of the 100nm VO₂ thin-film.

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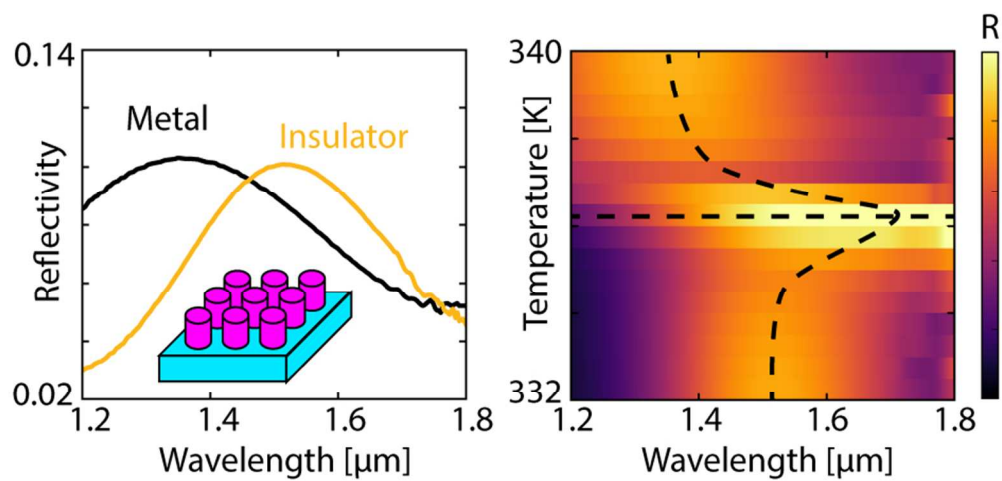
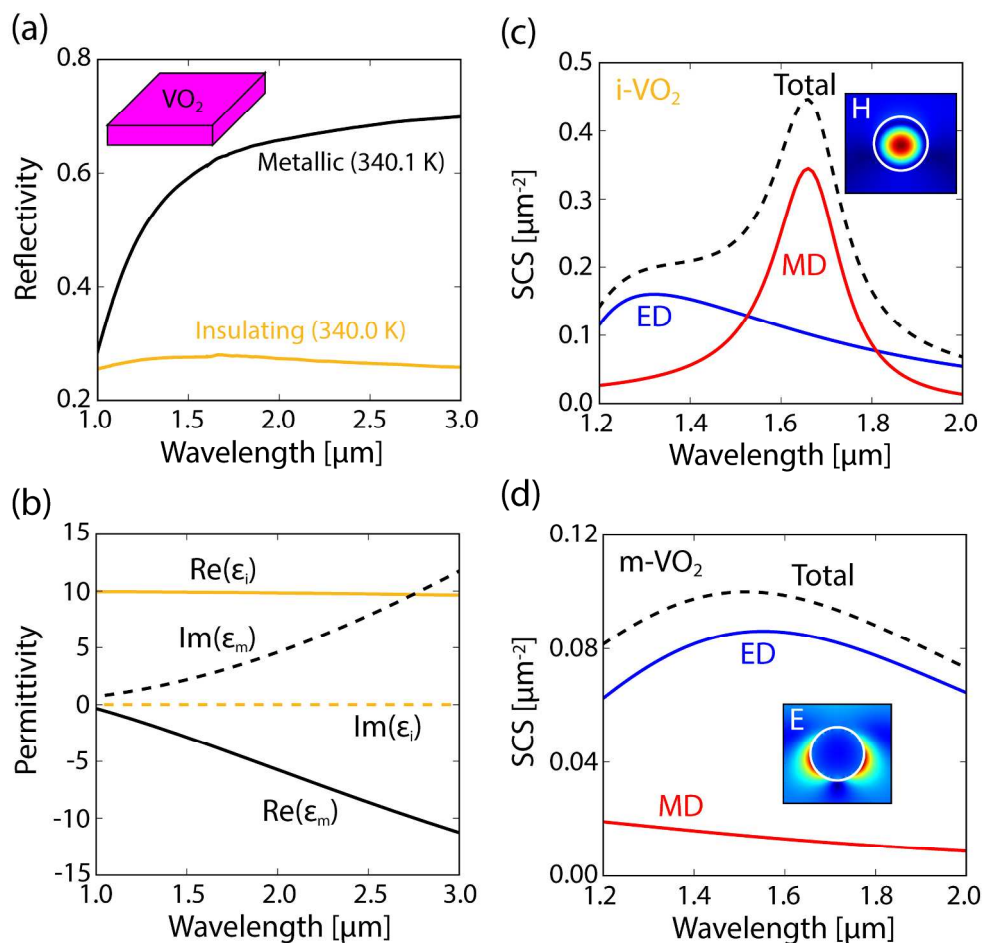


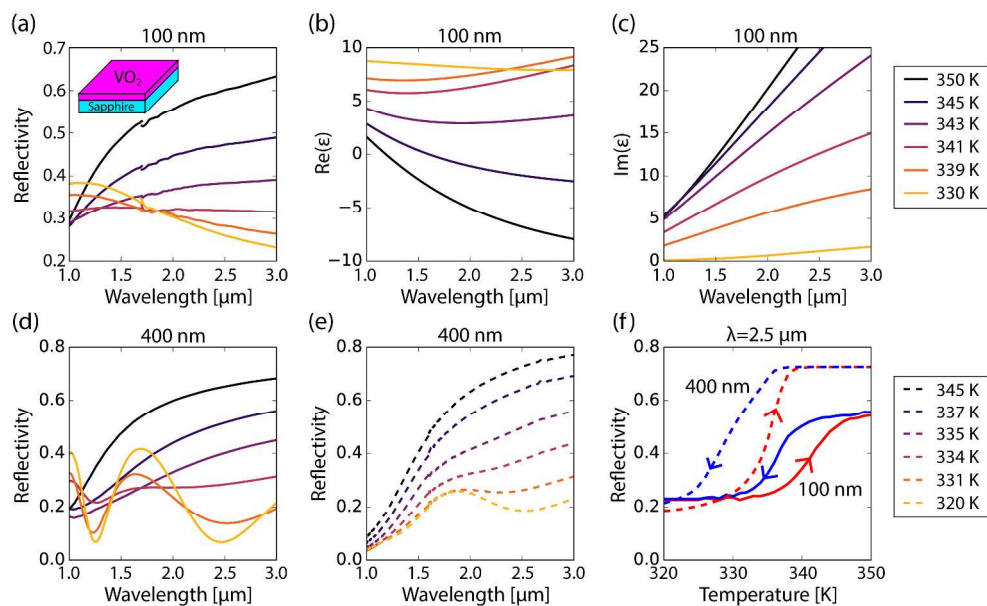
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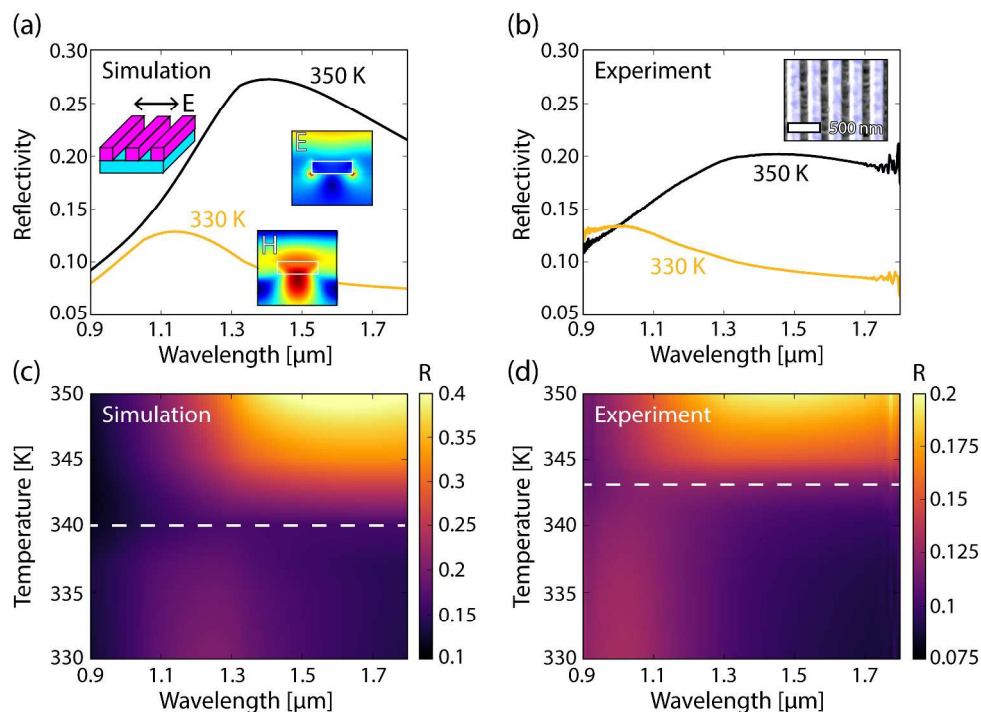
Reflectivity of VO₂ Single Crystals. (a) Near-infrared reflectivity of a VO₂ single crystal in its low-temperature insulating phase (orange curve) and high-temperature metallic phase (black curve). (b) Real (solid curve) and imaginary (dashed curve) components of the fitted VO₂ optical permittivity in the insulating phase (orange curve) and metallic phase (black curve). Scattering-cross-section of a (c) insulating, and (d) metallic VO₂ sphere in air, with diameter $d=500$ nm, calculated with Mie theory. Insets corresponds to simulated local (c) magnetic, and (d) electric, field profile at the fundamental resonance. Crystal growth, optical characterization, Mie Theory calculation, and simulation details are included in the Supporting Information.

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VO₂ Thin-Film Reflectivity and Optical Constants (a) The measured FTIR reflectivity of a 100 nm VO₂ film on a sapphire substrate. The (b) real, and (c) imaginary components of the optical permittivity, as determined by a spectral fit to the measured reflectivity. Experimental and fitting details are provided in the Methods and Supplementary Material. (d) The simulated reflectivity of a 400 nm thick VO₂ film, based on the 100 nm optical constants. (e) The measured reflectivity of a 400 nm thick VO₂ film. (f) The optical hysteresis of the 100 nm film (solid line) and 400 nm film (dashed line) at a wavelength $\lambda=2.5 \mu\text{m}$. Sample fabrication, measurement details, and mid-infrared spectra are included in the Supporting Information.

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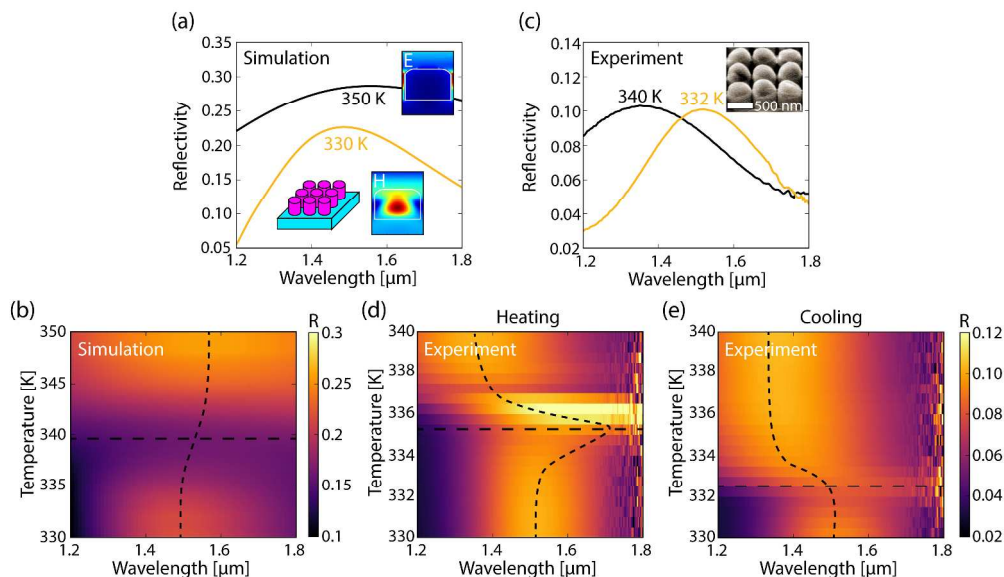


Figure 4 –VO₂ Disk Array Reflectivity. (a) Simulated reflectivity of a VO₂ disk array. Disk resonators have a height $h=500$ nm, diameter $d=480$ nm, and pitch $a=500$ nm based on optical constants from a 100 nm film. Insets correspond to local field distributions at the resonant peaks. (b) Simulated reflection as a function of temperature and wavelength. Vertical dashed curve tracks the reflectivity peak, horizontal dashed line represents the transition temperature (c) Experimentally measured reflectivity of VO₂ disk array (with approximate height $h=400$ nm, diameter $d=480$ nm, and pitch $a=500$) above and below transition. Inset is an SEM of the fabricated structure. (d) Experimental reflection data, upon heating, including intermediate temperatures. (e) Experimental reflection data, upon cooling, including intermediate temperatures. Vertical dashed curve tracks the reflectivity peak, horizontal dashed line represents the transition temperature. Note that the temperature span in (d,e) is different than (c). Simulation and experimental details are included in the Supporting Information.

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